

## Valence-band changes in $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$ and $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$ by transport and Shubnikov–de Haas effect measurements

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Measurements of galvanomagnetic effects in the temperature range 4.2–300 K and photoinduced “transient thermoelectric effect” (TTE) along the  $C_2$  axis at 300 K have been made for two types of solid solutions of semiconductors  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $0 \leq x \leq 0.4$ ) and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $0 \leq y \leq 1.8$ ). By incorporating In atoms into the  $\text{Sb}_2\text{Te}_3$  lattices, Hall coefficients, Hall mobilities, and the frequencies of Shubnikov–de Haas (SdH) oscillations are varied systematically. For  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$ , the Hall mobility is decreased with  $y$  up to  $y=0.7$  and then increased appreciably in the range  $0.7 < y < 1.8$ , and a frequency component of SdH oscillations is observed for  $y \geq 0.25$ . The observed TTE voltages decay exponentially with time, showing a multirelaxation process with characteristic relaxation times  $\tau_i$  ( $i=1,2,\dots$ ) for thermal diffusions of photoinduced conduction carriers, whose analyses give valuable information about carrier mobilities and effective masses. In the host material  $\text{Sb}_2\text{Te}_3$ , four relaxation times  $\tau_i$  ( $i=1-4$ ) are found, which are attributable to holes in the anisotropic upper and lower valence bands with effective-mass anisotropies of about 3. In addition, we have found two kinds of extra relaxation times  $\tau_i$  ( $i=5$  and  $6$ ) for  $y > 0.6$  in  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$ , confirming the existence of a valence band, whose anisotropy in the effective mass along the  $C_2$  direction is evaluated to be of the order of 2–2.5. Based on these experimental data we have proposed the most probable band model for these solid solutions.

### I. INTRODUCTION

Antimony telluride  $\text{Sb}_2\text{Te}_3$  belongs to a layered compound with the same tetradymite structure as  $\text{Bi}_2\text{Te}_3$ , in which a weak bond between Sb and Te layers exists. It usually involves antistructural defects (the occurrence of Sb atoms occupying Te lattice positions). Due to such native defects,  $\text{Sb}_2\text{Te}_3$  always shows a  $p$ -type conductivity with the hole concentration up to  $10^{20} \text{ cm}^{-3}$ . The upper valence band (UVB) of  $\text{Sb}_2\text{Te}_3$  consists of six ellipsoids tilted to the basal plane<sup>1,2</sup> (Drabble-Wolf model)<sup>3</sup> and the lower valence band (LVB), which is known to be a multivalley, but the anisotropy of this valley is unknown; there is a publication about a single valley model for LVB.<sup>4</sup> The value of the tilt angle  $\theta$  is not known exactly, but  $\theta \cong 50^\circ$  according to the extrapolation to  $x=1$  in  $(\text{Bi}_{1-x}\text{Sb}_x)_2\text{Te}_3$ .<sup>2</sup> The energy separation between the tops of the UVB and LVB is reported to be 150 meV,<sup>1</sup> but this value seems to be quite large; it may be of the order of 20–30 meV, because that for  $(\text{Bi}_{1-x}\text{Sb}_x)_2\text{Te}_3$  ( $0 < x < 1$ ) is about 30 meV.<sup>2</sup> Furthermore,  $\text{Sb}_2\text{Te}_3$  based solid solutions have been studied;<sup>5–9</sup> for example, the plasma fre-

quency at room temperature,<sup>5</sup> the lattice parameter and conductivity at room temperature<sup>6</sup> are known for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$ . According to optical studies at room temperature, the energy gap between UVB and lower electron band is increased by substituting In for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  mixed crystals.<sup>7,8</sup>

For further understanding of  $\text{Sb}_2\text{Te}_3$  based solid solutions, in the present work, we have carried out measurements of dc transport quantities in the temperature range 4.2–300 K, Shubnikov–de Haas (SdH) effect at 4.2 K in magnetic fields up to 35 T, and of pulsed laser induced “transient thermoelectric effect” (TTE) for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $0 \leq x \leq 0.4$ ) and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $0 \leq y \leq 1.8$ ) single crystals along the  $C_2$  axis at 300 K. The TTE method is a dynamic technique for understanding transport properties of multicarrier systems of electrons and holes, as evidenced for silicon,<sup>10</sup> GaAs,<sup>11</sup> quasi-two-dimensional charge-density-wave material  $\eta\text{-Mo}_4\text{O}_{11}$ ,<sup>12</sup> semimetal Bi,<sup>13</sup> and narrow gap semiconductor  $\text{Bi}_{2-x}\text{Sn}_x\text{Te}_3$ .<sup>14</sup> We are interested in the effect of the In or Se substitution on the electronic properties of the host  $\text{Sb}_2\text{Te}_3$  and getting valuable information about the changing of its energy spectrum, in particular, the valence bands.

## II. EXPERIMENT

$\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  single crystals were grown by a modified Bridgman method from 5N purity elements. The starting elements in the stoichiometric ratios were synthesized in evacuated conical silica ampoules. The crystals were easily cleaved perpendicular to the trigonal  $C_3$  axis, i.e., along the (0001) (or in the  $C_1C_2$ ) plane. The In and Se contents in the mixed crystals were determined by x-ray microprobe analyses. Samples for transport measurements were cut from cleaved single-crystal plates by a spark erosion machine (typical dimensions:  $5 \times 1.5 \times 0.4 \text{ mm}^3$ ).

For the investigation of the SdH effect, we used the experimental facilities with a pulse magnet available up to 35 T, at the University of Amsterdam. The resistivity and Hall effect were measured by a potentiometric method with the dc along the  $C_2$  direction in magnetic fields up to 7 T applied along the  $C_3$  axis (perpendicular to the layers), using a superconducting solenoid. The experimental setup and measuring principle of the TTE method have been described earlier.<sup>10–14</sup> In principle, a pulsed laser (laser power  $\sim 350 \text{ mJ}$ ) produced by a Nd:YAG laser source with the wavelength of 1060 nm ( $=1.17 \text{ eV}$ ) and pulse width of 8 ns was irradiated normal to one end of crystal. Photoinduced TTE voltages were detected over the wide time range 50 ns to 500 ms by a digital storage oscilloscope through a homemade preamplifier, whose output signal was fed to a computer for record and numerical analysis. TTE measurements were made only at 300 K, below which the TTE signals became too small to be detectable.

## III. EXPERIMENTAL RESULTS

### A. Galvanomagnetic effect and Shubnikov–de Haas effect

Figures 1(a) and 1(b) show the temperature dependence of the resistivity  $\rho$  for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $x=0.02, 0.1, 0.2$ , and  $0.4$ ) and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $y=0, 1.0, 1.6$ , and  $1.8$ ) along the  $C_2$  direction in logarithmic scales, respectively. For all samples,  $\rho$  decreases with decreasing temperature and then approaches a constant value at low temperatures. Residual resistivities increase with the substitution of In or Se atoms, which are due to the increased ionized or neutral impurity scatterings. In the temperature range 77–300 K, the  $\rho$ – $T$  curves for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  obey a power law of the form  $\rho = T^n$ , with the exponent  $n \approx 1.3$  for all samples.

The Hall coefficient  $R_H$  is positive for all samples and is almost independent of temperature in the whole temperature range 4.2–300 K and of a magnetic field up to 7 T. The values of  $R_H$  and the Hall mobilities  $\mu_H$  at 4.2 K are plotted against the In or Se content for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  in Figs. 2(a) and 2(b), respectively. The Hall coefficient increases monotonically with increasing the concentration of In or Se, which indicates that the apparent hole concentrations decrease with  $x$  or  $y$ . On the other hand, the Hall mobility  $\mu_H$  decreases drastically with increasing  $x$  for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$ , while for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  it falls at a low Se concentration ( $y \leq 0.7$ )

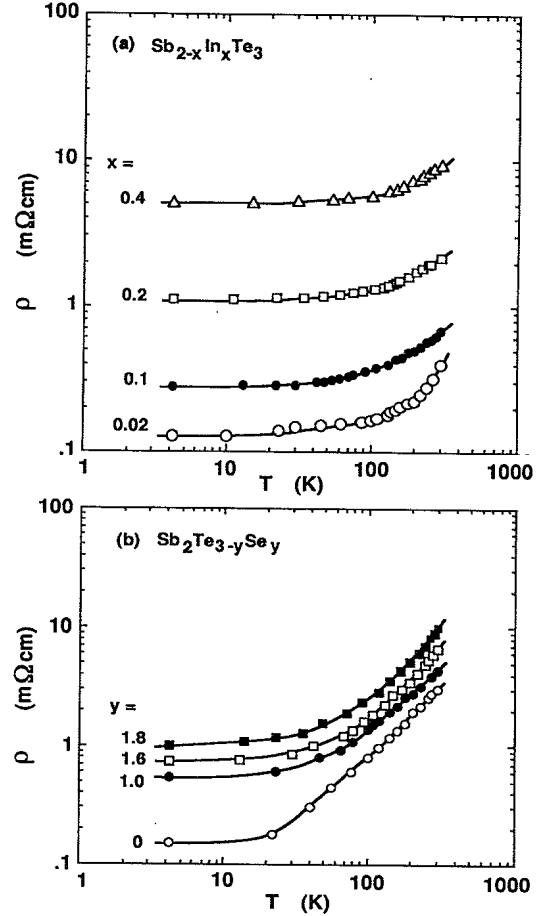


FIG. 1. Temperature dependence of the resistivity along the  $C_2$  axis for (a)  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $0 \leq x \leq 0.4$ ) and (b)  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $0 \leq y \leq 1.8$ ).

and then increases up to  $y = 1.8$  (which becomes higher than that for the host material), followed by a decrease. The drastic decrease in  $\mu_H$  with  $x$  for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  may be due to an appreciable increase of vacancies, as found for  $\text{In}_2\text{Te}_3$  or  $\text{In}_2\text{Se}_3$  crystals.<sup>15,16</sup> For the  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  system, the increase in  $\mu_H$  for  $y > 0.7$  is due to the predominance of a valence band with a smaller effective mass compared to those of the main UVB and LVB, as described later.

The magnetoresistance of these crystals is proportional to the square of a magnetic field  $B$  at a low magnetic fields, while it shows SdH oscillations at high fields. Figure 3 shows typical results for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $x=0, 0.02$ , and  $0.2$ ) at 4.2 K. In the present orientation of  $B$  parallel to the  $C_3$  axis, we have observed a single frequency, which indicates that the extremal cross sections of the six ellipsoids of the Fermi surfaces coincide. Due to a very sharp decrease in the Hall mobility with increasing In content of  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$ , the oscillation amplitude falls rapidly with  $x$ ; even for  $x=0.2$ , it was impossible to detect the SdH effect in magnetic fields up to 35 T.

Figure 4 shows similar SdH signals at 4.2 K for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$ , with Se concentrations up to  $y = 1.8$ , where we see that a beating in the SdH oscillations appears for

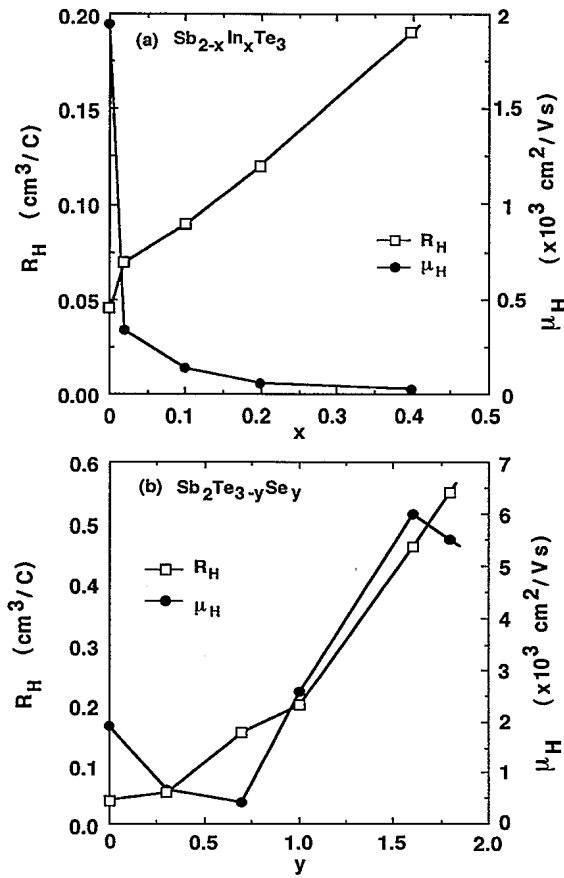


FIG. 2. Hall coefficients  $R_H$  and Hall mobilities  $\mu_H$  at 4.2 K plotted against (a) In content  $x$  for  $Sb_{2-x}In_xTe_3$  and (b) Se content  $y$  for  $Sb_2Te_{3-y}Se_y$ .

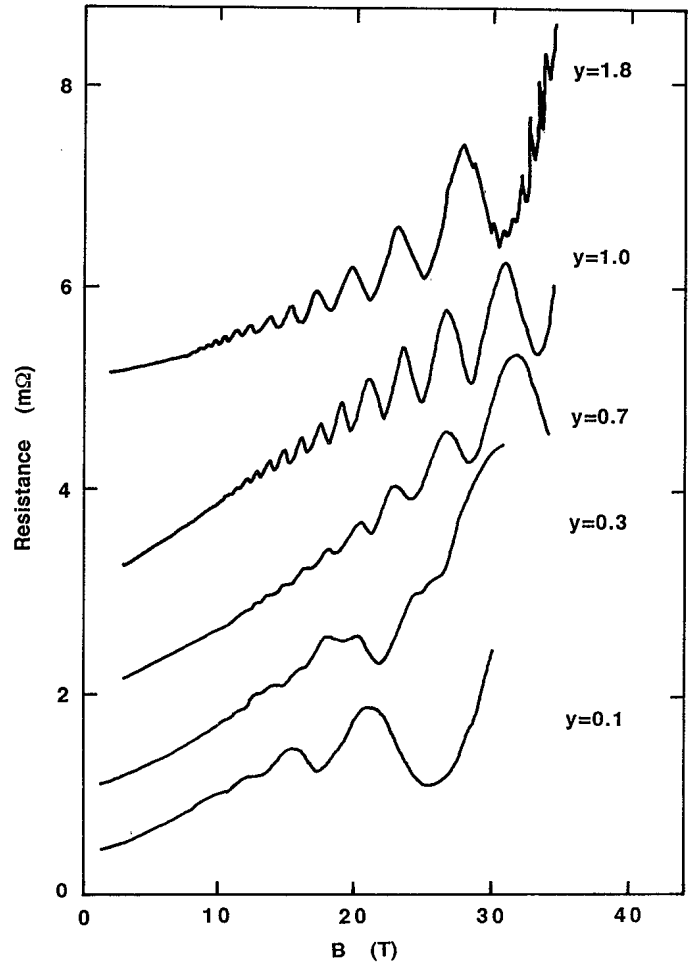


FIG. 4. SdH oscillations under the magnetic field  $B$ , parallel to the  $C_3$  axis for  $Sb_2Te_{3-y}Se_y$  ( $y = 0.1, 0.3, 0.7, 1.0,$  and  $1.8$ ) at 4.2 K.

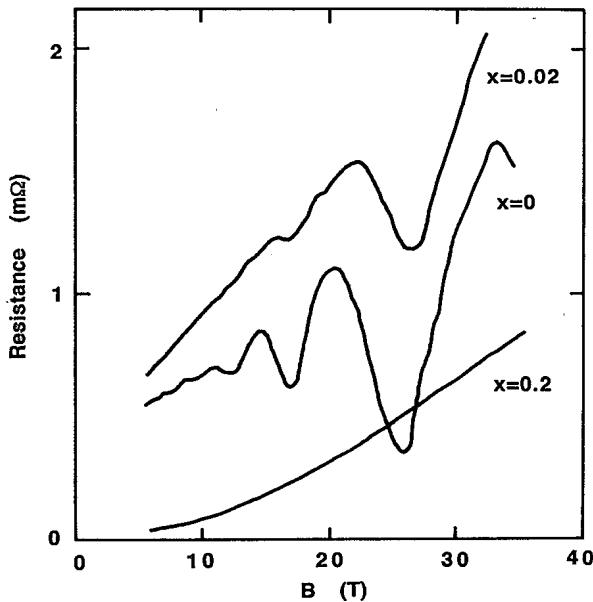


FIG. 3. SdH oscillations under the magnetic field  $B$ , parallel to the  $C_3$  axis for  $Sb_{2-x}In_xTe_3$  ( $x = 0, 0.02,$  and  $0.2$ ) at 4.2 K.

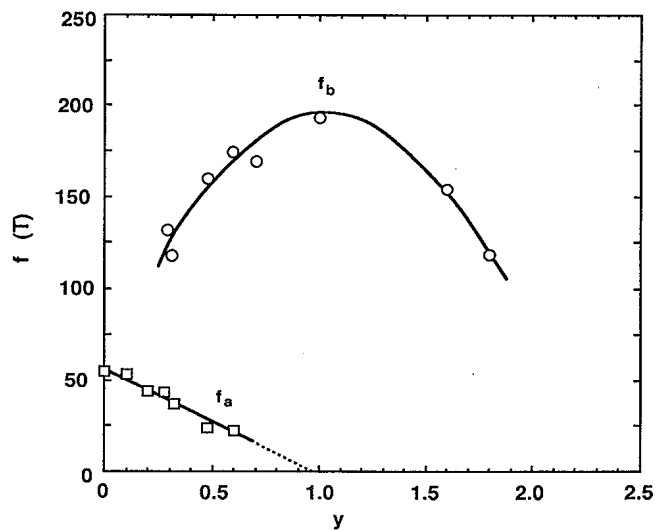


FIG. 5. SdH oscillation frequencies  $f_a$  and  $f_b$  at 4.2 K plotted against Se content  $y$  for  $Sb_2Te_{3-y}Se_y$ .

$y > 0.2$ , indicating the presence of a different frequency. The Fourier transforms of the SdH oscillations give us the frequency components, which are plotted against Se concentration  $y$  in Fig. 5. The main frequency  $f_a$ , which corresponds to the extremal cross section of the Fermi surfaces of the UVB of  $\text{Sb}_2\text{Te}_3$ , falls with increasing  $y$  (which indicates that the cross-sectional area of the UVB is decreased with the Se content), and disappears near  $y = 1.0$  according to the extrapolation. A frequency component  $f_b$  appears for samples with  $y > 0.2$ , which increases up to  $y = 1.0$ , followed by a decrease. This frequency can be ascribable to a valence band (thereafter denoted by NVB), as it will be discussed later. In addition, we have found a much higher-frequency component  $f_c = 1600$  T for sample with  $y = 1.8$ . Using these data, together with the following TTE data, we shall discuss the possible band model for these solid solutions.

### B. Transient thermoelectric effect

Figure 6 shows typical photoinduced TTE signals along the  $C_2$  direction for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $x = 0$  and  $0.2$ ) and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $y = 0.2, 0.5$ , and  $1.0$ ) at  $T = 300$  K in the time intervals  $0-20 \mu\text{s}$ , where the observed TTE signals are averaged out by 15–20 times, because of extremely small magnitudes compared to those of the similar family of  $\text{Bi}_{2-x}\text{Sn}_x\text{Te}_3$  crystals.<sup>14</sup> For the host ma-

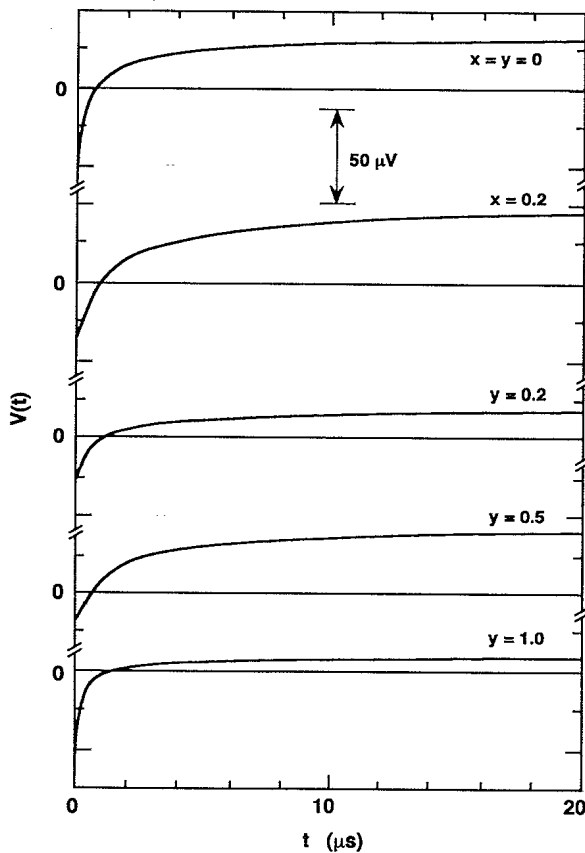


FIG. 6. Typical photoinduced TTE voltages  $V(t)$  for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $x = 0, 0.2$ ) and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $y = 0.2, 0.5$ , and  $1.0$ ) along the  $C_2$  axis at 300 K.

terial  $\text{Sb}_2\text{Te}_3$ , immediately after the laser irradiation, the induced TTE voltage  $V(t)$  decreases drastically within 50 ns (minimum sampling time) and increases with time  $t$  up to about  $25 \mu\text{V}$  in the short-time range  $0-20 \mu\text{s}$ . Similar TTE profiles are also observed for the solid solutions.

As in the case for various solids,<sup>10–14</sup> the TTE voltages  $V(t)$  at time  $t$  can be expressed in the exponential form,

$$V(t) = V_0 + \sum a_i \exp(-t/\tau_i), \quad (1)$$

where  $V_0$  is a constant value at  $t \rightarrow \infty$ ,  $a_i$  a relaxation amplitude, and  $\tau_i$  a relaxation time for the  $i$ th relaxation process or the type of carriers;  $a_i > 0$  or  $a_i < 0$  corresponds to a thermal diffusion of electrons or holes, respectively. From the analyses of the observed TTE signals, we have obtained four kinds of relaxation times  $\tau_i$  ( $i = 1-4$ ), due to “holes” ( $a_i < 0$ ) for the host material and for In-substituted samples ( $x \leq 0.2$ ). For Se-substituted ones, we have also four relaxation times  $\tau_1 - \tau_4$ , but the TTE signal corresponding to  $\tau_1$  becomes too weak for higher Se concentration  $y > 1.0$ . Moreover, two extra relaxation times  $\tau_i$  ( $i = 5$  and  $6$ ;  $a_i < 0$ ) are observed for a higher Se concentration  $y \geq 0.7$ . Figures 7(a) and 7(b) plot the values of  $\tau_i$  for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  and

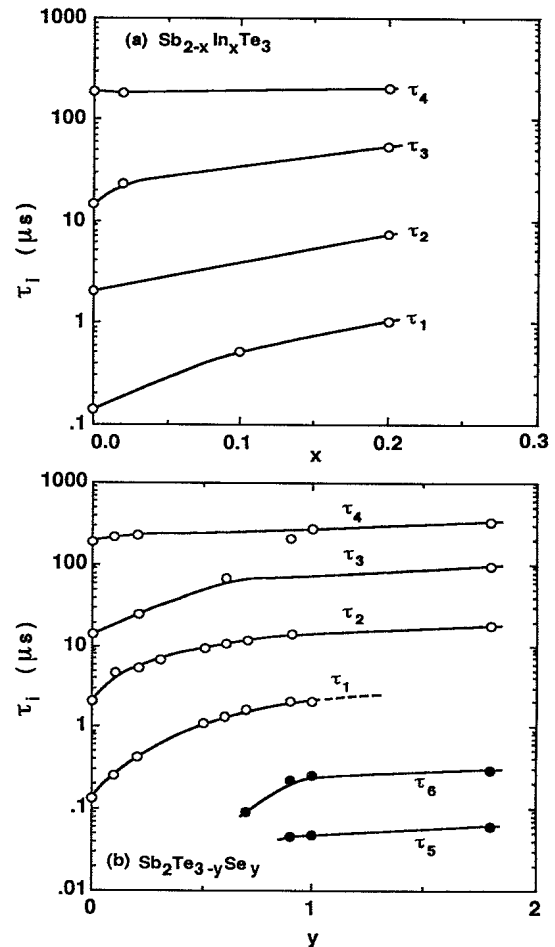


FIG. 7. (a) Observed relaxation times  $\tau_i$  ( $i = 1-4$ ) for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $0 \leq x \leq 0.2$ ) plotted against In content  $x$  and (b) those of  $\tau_i$  ( $i = 1-6$ ) for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $0 \leq y \leq 1.8$ ) plotted against Se content  $y$ .

$\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  against  $x$  and  $y$  in semilogarithmic scales, respectively, where we see that these values span a wide time range from  $10^{-2}$  to  $10^2 \mu\text{s}$ , and that all relaxation times increase with  $x$  or  $y$ .

#### IV. DISCUSSIONS

##### A. $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$

It is known that the UVB of the host  $\text{Sb}_2\text{Te}_3$  consists of six equivalent valleys, similar to that of the same family of  $\text{Bi}_2\text{Te}_3$ , whose Fermi surfaces have an ellipsoidal shape tilted by about  $50^\circ$  from the  $C_1C_2$  basal plane (perpendicular to the  $C_3$  direction),<sup>1,2</sup> as shown schematically in Fig. 8(Ia), where the top view of the schematic Fermi surfaces of the UVB in the  $C_1C_2$  plane is illustrated in the Brillouin zone. In fact, our observed SdH oscillation for  $\text{Sb}_2\text{Te}_3$  shows only the single frequency component  $f_a$ , indicating that all the Fermi surfaces are equivalent in the  $C_3$  direction. In addition, arrows mark the two valleys with different effective masses along the  $C_2$  direction, to which we attribute the observed relaxation times  $\tau_1$  and  $\tau_2$  of the TTE signals, as indicated. On the other hand, much less is known about the Fermi surface of the LVB. In order to explain our experimental results reasonably, we propose a six-valley model for the LVB, as in the case for  $\text{Bi}_2\text{Te}_3$ ,<sup>14</sup> where we assign another two relaxation times  $\tau_3$  and  $\tau_4$  to two different valleys of the LVB along the  $C_2$  direction, similar to the case of the UVB [Fig. 8(Ia)]; however, its location in the Brillouin zone is unknown. The schematic band structure of  $\text{Sb}_2\text{Te}_3$  is shown in Fig. 8(Ib), with which we shall give discussions about transport parameters (see later).

Now according to the ellipsoidal nonparabolic model,<sup>2,3</sup> the energy spectrum of the holes in the UVB of  $\text{Sb}_2\text{Te}_3$  is represented in the form

$$2m_0E = h^2(\alpha_{11}k_1^2 + \alpha_{22}k_2^2 + \alpha_{33}k_3^2 + 2\alpha_{23}k_2k_3), \quad (2)$$

where the inverse mass tensor components  $\alpha_{ij}$  depend on

the energy. Here, the momentum  $k_1$ ,  $k_2$ , and  $k_3$  form the Cartesian coordinates, which are parallel to one of the binary and bisectrix axes, and to the trigonal axis, respectively. The period of SdH oscillations  $\Delta(1/B)$  in this model is connected with these parameters, as

$$\Delta \left( \frac{1}{B} \right) = \frac{eh}{m_0 E_F} [(\alpha_{22}\alpha_{33} - \alpha_{23}^2)\cos^2\alpha + \alpha_{11}\alpha_{33}\cos^2\beta + \alpha_{11}\alpha_{22}\cos^2\gamma + 2\alpha_{11}\alpha_{23}\cos\beta\cos\gamma]^{1/2}. \quad (3)$$

Here,  $\cos\alpha$ ,  $\cos\beta$ , and  $\cos\gamma$  are the direction cosines of the magnetic field  $B$  relative to the  $k_i$  axis ( $i=1,2,3$ ) in the momentum space, respectively. We have calculated the hole concentrations  $p_{\text{SdH}}$  at 4.2 K, for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$ , using the parameters of the energy spectrum for  $\text{Sb}_2\text{Te}_3$ ,<sup>2</sup> where we have assumed that the anisotropy of the Fermi surface and tilt angle of the ellipsoids do not depend on the In concentration. These values, together with those evaluated from the Hall effect  $p_H$ , are compared in Table I. We note that with increasing In content the hole concentration is decreased, which may due to the suppression of antistructural defects formed in the host material. In addition,  $p_H$  is larger than  $p_{\text{SdH}}$  in each sample, which means that the holes in the UVB with  $p_{\text{SdH}}$  are responsible for the SdH effect, while those in the LVB are not, due to very low mobility; thus,  $p_{\text{SdH}}$  is regarded as the hole concentration  $p_U$  in the UVB.

On the other hand, the theoretical analysis of the diffusion equation for photogenerated carriers gives the following relation between the relaxation time  $\tau_i$  and carrier mobility  $\mu_i$  of the  $i$ th carrier:<sup>10</sup>

$$\tau_i = eL_i^2 / (2k_B T \mu_i) \quad \text{or} \quad \mu_i = eL_i^2 / (2k_B T \tau_i), \quad (4)$$

where  $L_i$  is a diffusion length, which, we assume, is independent of the type of carriers ( $L_i = L$ ).<sup>10-14</sup> Based on the above two-valence-band model, we have evaluated the hole concentration  $p_U$  in the UVB,  $p_L$  in the LVB, and their mobility  $\mu_i$  along the  $C_2$  axis corresponding to  $\tau_i$

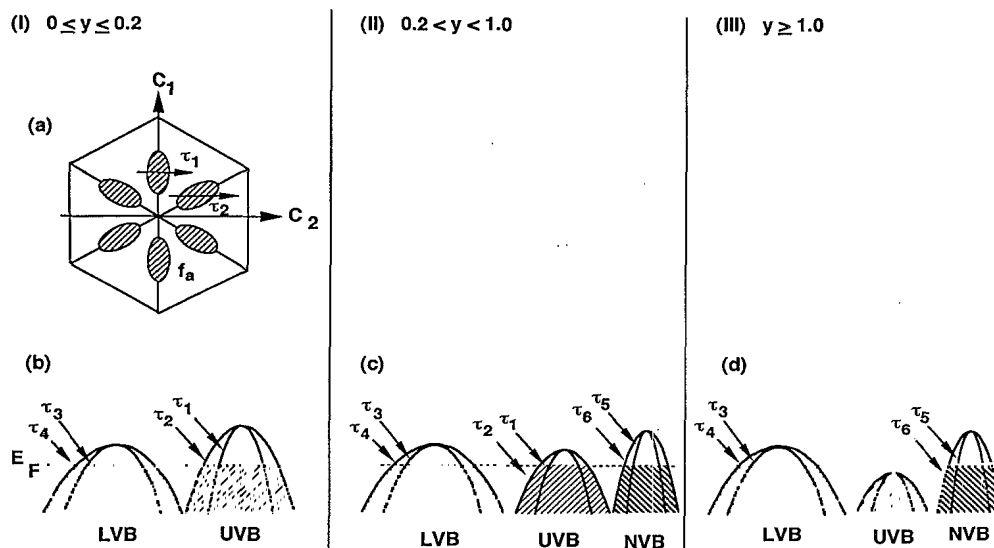


FIG. 8. Schematic valence-band model for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  in different Se concentration regions; (I)  $y \leq 0.2$ , (II)  $0.2 < y < 1.0$ , and (III)  $y \geq 1.0$  (see text).

TABLE I. Hole concentration  $p_H$  evaluated from the Hall coefficient,  $p_{\text{SdH}}$  from the SdH effect (which is assumed to be equal to the hole concentration  $p_U$  in the UVB),  $p_L$  in the LVB, parameters  $\alpha$  and  $\beta$ , hole mobilities  $\mu_1$  and  $\mu_3$ , and effective-mass ratios  $m_2^*/m_1^*$  and  $m_4^*/m_3^*$  evaluated using the two-valence-band model for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $x \leq 0.2$ ) (see text).

$x$	$p_H$	$p_{\text{SdH}}=p_U$ ( $10^{20} \text{ cm}^{-3}$ )	$p_L$	$\alpha$ ( $10^{-2}$ )	$\beta$ ( $10^2$ )	$\mu_1$ ( $\text{cm}^2/\text{V s}$ )	$\mu_3$	$m_2^*/m_1^*$	$m_4^*/m_3^*$
0	1.55	0.23	90	0.93	3.9	8000	81	2.9	2.8
0.02	0.89	0.20	80	0.78	4.0	1400	11		2.3
0.1	0.69	0.18	41	1.3	2.3	560	7.3		
0.2	0.52							2.2	1.7

using Eq. (4), as in the following way. Here, it is to be noted that to the carrier transport along the  $C_2$  direction, there contribute two equivalent hole pockets corresponding to  $\tau_1$  (or carrier mobility  $\mu_1$ ) and four corresponding to  $\tau_2$  (or  $\mu_2$ ) in the UVB, as shown in Fig. 8(Ia). Similarly, for the LVB we propose two equivalent pockets corresponding to  $\tau_3$  (or  $\mu_3$ ) and four ones with  $\tau_4$  (or  $\mu_4$ ). Also, we have taken account of the inequalities for these quantities;  $\tau_1 \ll \tau_2$  (or  $\mu_1 \gg \mu_2$ ) and  $\tau_3 \ll \tau_4$  (or  $\mu_3 \gg \mu_4$ ).

Thus, the total conductivity  $\sigma$ , Hall coefficient  $R_H$ , and Hall mobility  $\mu_H$  can be written as

$$\sigma = e \left\{ \left(\frac{2}{6}\right) p_U \mu_1 + \left(\frac{4}{6}\right) p_U \mu_2 + \left(\frac{2}{6}\right) p_L \mu_3 + \left(\frac{4}{6}\right) p_L \mu_4 \right\} \\ \cong (e/3) (p_U \mu_1 + p_L \mu_3) = (e/3) p_U \mu_1 (1 + \alpha \beta), \quad (5)$$

$$R_H = (1/e) \left\{ \left(\frac{2}{6}\right) p_U \mu_1^2 + \left(\frac{4}{6}\right) p_U \mu_2^2 + \left(\frac{2}{6}\right) p_L \mu_3^2 + \left(\frac{4}{6}\right) p_L \mu_4^2 \right\} \\ \times \left\{ \left(\frac{2}{6}\right) p_U \mu_1 + \left(\frac{4}{6}\right) p_U \mu_2 + \left(\frac{2}{6}\right) p_L \mu_3 + \left(\frac{4}{6}\right) p_L \mu_4 \right\}^{-2} \\ \cong (3/ep_U) (1 + \alpha^2 \beta) / (1 + \alpha \beta)^2, \quad (6)$$

$$\mu_H \cong \mu_1 (1 + \alpha^2 \beta) / (1 + \alpha \beta), \quad (7)$$

with  $\alpha = \mu_3/\mu_1$  ( $=\tau_1/\tau_3$ ) and  $\beta = p_L/p_U$  (electron charge  $e > 0$ ). Taking into account the experimental fact that  $R_H$  is almost independent of temperature, we assume that the parameters  $\alpha$  and  $\beta$  in Eqs. (5)–(7) are also temperature independent. According to Eq. (4), the mobility ratio  $\alpha$  is expressed by the ratio for the relaxation times of the UVB and LVB,  $\tau_1$  and  $\tau_3$ , respectively, measured along the  $C_2$  axis [Fig. 7(a)]. The hole concentration  $p_U$  in the UVB is regarded as equal to the observed value  $p_{\text{SdH}}$  from the SdH effect, using Eq. (3) with parameters that were determined for  $\text{Sb}_2\text{Te}_3$ .<sup>2</sup> The total hole concentration  $p_H = 1/eR_H$  was calculated from the observed Hall coefficient [Fig. 2(a)]. With the values of  $p_U$  and  $\alpha$ , we get the parameter  $\beta$  using Eq. (6), from which we obtain the hole concentration  $p_L$  in the LVB. The observed Hall mobility  $\mu_H$  is further approximated by  $\mu_H = \mu_1/(1 + \alpha \beta)$  (since  $\alpha^2 \beta \ll 1$ ) in Eq. (7), with which we obtain the value of  $\mu_1$  and thus the carrier mobility  $\mu_3$  for the LVB from  $\alpha$ . These evaluated parameters are compiled in Table I. The hole concentrations  $p_U$  and  $p_L$  are decreased with increasing In content, primarily due to the suppression of the antistructural defects by incorporating of In atoms into the Sb lattices. We should also note that the hole mobility  $\mu_1$  of the UVB is much larger than  $\mu_3$  of the LVB, and these values are drastically reduced with  $x$ , which indicates that ionized or neutral im-

purity scatterings by these crystal defects become predominant.

Moreover, the substitution of In atoms into Sb sites in  $\text{Sb}_2\text{Te}_3$  may affect the anisotropy of the ellipsoidal Fermi surfaces of the UVB and LVB. To estimate the anisotropy in the effective mass, we use the following relation:

$$\tau_j/\tau_i = \mu_i/\mu_j = (m_j^*/m_i^*)^\gamma, \quad (8)$$

where  $\gamma$  is a parameter characterizing a scattering mechanism. Assuming phonon scatterings for all carriers in each pocket ( $\gamma = \frac{5}{2}$ ), from Eq. (8) with the observed relaxation times at 300 K [Fig. 7(a)], we obtain the effective mass ratio with respect to  $m_1^*$  for the UVB,  $m_2^*/m_1^*$ , and that for the LVB with respect to  $m_3^*$ ,  $m_4^*/m_3^*$ , as listed in Table I. We see that the anisotropy in the effective mass of the UVB is of the same order of magnitude as that of the LVB for the host material (about 3), and these values decrease with In alloying. In particular, the reduction is appreciable for the latter, which may arise from the increasing in the tilt angle  $\theta$  with In alloying.

### B. $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$

To our knowledge, there is no available information about the band parameters in Eq. (2) for the solid solution of  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$ . Hence, in the present work, based on our band model (Fig. 8), we have attempted to evaluate the hole concentrations  $p_U$  and  $p_L$ , and hole mobilities  $\mu_1$ ,  $\mu_3$ , and  $\mu_5$  of the UVB, LVB, and valence band, respectively, using the observed Hall data [Fig. 2(b)], the SdH frequencies  $f_a$  and  $f_b$  (Fig. 5), and TTE data  $\tau_i$  [Fig. 7(b)]. We should note that there are three different regions of Se concentration: In the first region  $0 \leq y \leq 0.2$ , there is only one frequency component in the SdH effect, but the Hall concentration is higher than the SdH concentration, and we have observed four relaxation times. In the second region  $0.2 < y < 1.0$ , there are two frequency components  $f_a$  and  $f_b$ , and two additional relaxation times  $\tau_5$  and  $\tau_6$  are detected for  $y \geq 0.7$ , which we have attributed to a different valence band, with an anisotropic and small effective mass, as discussed later. The frequency  $f_b$  increases with  $y$  up to  $y = 1.0$  and then decreases (Fig. 5). The increase in the frequency with  $y$  in the range  $0.2 < y < 1.0$  is due to the change of the positions of the UVB, LVB, and valence band; the valence band moves up and UVB moves down, due to Se doping [(Fig. 8(c)]. For  $y \geq 1.0$ , there are two extremes filled by holes, valence band and LVB, because UVB disappears according to the

extrapolation (Fig. 5).

In view of these, we propose the most likely band structure for the solid solutions with three regions,  $0 \leq y \leq 0.2$ ,  $0.2 < y < 1.0$ , and  $1.0 \leq y \leq 1.8$ , as depicted schematically in Figs. 8(I), 8(II), and 8(III), respectively. The Fermi surfaces in the  $C_1C_2$  plane for the UVB consist of two types of hole valleys with the relaxation times  $\tau_1$  and  $\tau_2$ , as marked by arrows in Fig. 8(Ia). We assume that the LVB is the same as that of the host [Fig. 8(Ia)], that has a multivalley structure consisting of two types of hole valleys with the relaxation times  $\tau_3$  and  $\tau_4$  in the whole region of Se concentration.

With this model, we have evaluated the carrier concentration  $p_U$ , for the UVB in the region I from the observed SdH frequency  $f_a$ , using Eq. (3) with the parameters that are extrapolated from those for  $\text{Sb}_2\text{Te}_3$ .<sup>2</sup> Then using Eqs. (5)–(8) with the experimental values of  $\alpha$  and Hall-effect data, the values of  $\beta$  can be evaluated according to the procedure similar to that for  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$ . All calculated values are listed in Table II. We note that the carrier mobilities are decreased with  $y$  drastically, which may be due to an appreciable increase of ionized and/or neutral scattering centers by substituting Se atoms into Te lattices.

However, this procedure is not acceptable for  $0.2 < y < 1.0$ , because in this region II, we have simultaneously three valence bands, UVB, LVB, and valence band [Fig. 8(II)], where the valence band is supposed to consist of six ellipsoids with a small anisotropy compared to that of the UVB. With such a three-valence-band model, the Hall coefficient  $R_H$  and the Hall mobility  $\mu_H$  are written as

$$R_H = (3/ep_U)(1 + \alpha^2\beta + \delta^2\epsilon + 2\gamma^2\epsilon)/(1 + \alpha\beta + \delta\epsilon + 2\gamma\epsilon)^2, \quad (9)$$

$$\mu_H = \mu_1(1 + \alpha^2\beta + \delta^2\epsilon + 2\gamma^2\epsilon)/(1 + \alpha\beta + \delta\epsilon + 2\gamma\epsilon), \quad (10)$$

respectively, where  $\alpha = \mu_3/\mu_1$ ,  $\beta = p_L/p_U$ ,  $\gamma = \mu_6/\mu_1$ ,  $\delta = \mu_5/\mu_1$ , and  $\epsilon = p_N/p_U$  ( $p_U$  is the hole concentration in the valence band). Here, we note that  $\mu_1 \gg \mu_3$  ( $\tau_1 \ll \tau_3$ ),

$\mu_3 \gg \mu_4$  ( $\tau_3 \ll \tau_4$ ), and  $\mu_5 \sim 5\mu_6$  [ $\tau_5 \sim (\frac{1}{5})\tau_6$ ; see Fig. 7(b)]; thus, in actual the terms,  $2\gamma^2\epsilon$  in the above expressions is negligibly small.

For the self-consistent fitting in this region II using Eqs. (2), (3), (9), and (10), we need simultaneously the static transport, SdH effect, and dynamic TTE data. The parameter  $\alpha$  can be readily obtainable from the observed relaxation times  $\tau_1$  and  $\tau_3$ , while  $\delta$  cannot be obtained because  $\tau_5$  is unknown in this region; we then extrapolate the data at the value at  $y = 0.7$  ( $\tau_5 = 0.035 \mu\text{s}$ ) from Fig. 7(b) to get the parameter  $\delta$  for  $y = 0.7$  ( $\delta = 45$ ). Since the parameter  $\beta$  in the region I is seen to be nearly constant [ $(2.1-3.7) \times 10^2$ ], we use the averaged value of  $\beta = 2.7 \times 10^2$  for the region II. On the other hand, the hole concentrations  $p_U$  are estimated from the values of  $f_a$ , using Eqs. (2) and (3) with the band parameters for  $(\text{Bi}_{1-x}\text{Sb}_x)_2\text{Te}_3$  ( $0 < x < 1$ );<sup>2</sup> we obtain a good relationship between  $f_a$  and  $p_U$ , as  $p_U \propto f_a^{3/2}$ , which is used also for the valence band, as  $p_N \propto f_b^{3/2}$  (see below). The value of  $f_a$  for  $y = 0.7$  is also evaluated to be 14 T from the extrapolation of the experimental data (Fig. 5). Thus, only for  $y = 0.7$ , have we obtained the best-fit parameters  $\epsilon$ ,  $\mu_1$ ,  $p_N$  ( $=\epsilon p_U$ ), and  $\mu_5$  ( $=\delta\mu_1$ ) for the valence band using Eqs. (9) and (10). To calculate  $p_N$  for other Se concentrations, furthermore, we have assumed that the anisotropy of the Fermi surfaces is constant in the region II, where we use the relation  $p_N \propto f_b^{3/2}$ . With the above procedure, Table II lists the available parameters in the region II.

Moreover, for the region III with  $y \geq 1.0$ , we have attempted to evaluate the parameters using the two-valence-band model and Eqs. (5)–(7), with the same approximation for the shape of the valence band as was done for  $y = 0.7$ . Here, we define  $\alpha' = \mu_3/\mu_5$  and  $\beta' = p_L/p_N$ . According to the procedure as done in the region I, we have evaluated these values, including the hole mobilities  $\mu_3$  and  $\mu_5$  at 300 K, as listed in Table II. We note that the carrier mobility of the valence band  $\mu_5$  in this region is particularly large. However, for  $y = 1.6$  and 1.8, it is difficult to fit all data under the assumption that the tilt angle and the shape of the ellipsoids of the

TABLE II. Best-fit parameters for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  in the Se concentration regions, (I)  $0 \leq y \leq 0.2$ , (II)  $0.2 < y < 1.0$ , and (III)  $y \geq 1.0$  (see text).

Region	$y$	$p_U$	$p_L$ ( $10^{19} \text{ cm}^{-3}$ )	$p_N$	$\alpha$ ( $10^{-2}$ )	$\alpha'$ ( $10^{-4}$ )	$\beta$ ( $10^2$ )	$\beta'$	$\delta$	$\epsilon$	$\mu_1$	$\mu_3$ ( $\text{cm}^2/\text{V s}$ )	$\mu_5$
(I)	0	2.43	900		0.93		3.7				8210	76	
	0.1	2.25	540		1.3		2.4				2630	34	
	0.2	1.70	360		1.6		2.1				1810	29	
(II)	0.25	1.64		4.7	1.8		(2.7)			2.9			
	0.3	1.34		3.9			(2.7)			2.9			
	0.5	0.73		6.3	1.8		(2.7)			8.6			
	0.6	0.60		7.1	1.8		(2.7)			12			
	0.7	0.30		6.6	2.3		(2.7)		45	22	12	0.3	540
(III)	1.0		570	7.0		7.0		81				2.0	2750
	1.6		120	4.0		7.2		29				5.7	7900
	1.8		140	3.2		7.5		43				5.5	7290

valence band are constant; if this is the case, we have to reduce  $p_N$  by 30%. Thus the anisotropy or tilt angle should change in the valence band ellipsoids by Se doping. Moreover, the ratio  $f_a/f_b$  from Fig. 5 is found to be approximately equal to  $\beta^{2/3}$ , from which we may ascribe the observed high frequency  $f_c$  (1,600 T) of the SdH effect for  $y = 1.8$  to the LVB. The hole concentrations  $p_U$  and  $p_L$  for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  are decreased with increasing  $y$ , indicating that the energy gap between the valence and conduction bands is increased, similar to that found for the  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  system.<sup>7,8</sup>

Finally, we have estimated an anisotropy in the effective-mass ratio  $m_i^*/m_j^*$  for each band, using Eq. (8) with the observed relaxation times for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  along the  $C_2$  axis;  $m_2^*/m_1^* = 3-2$  for UVB and  $m_4^*/m_3^* = 3-1.5$  for LVB in the range  $0 \leq y \leq 1.8$ , indicating that the anisotropies for these bands are decreased by substituting Se atoms into the host  $\text{Sb}_2\text{Te}_3$ , which may be due to the increase in the tilt angle of the ellipsoidal Fermi surface, with respect to the basal plane of each band, rather than the change in its size. On the other hand, the ratio  $m_6^*/m_5^*$  for the valence band in the range  $0.9 \leq y \leq 1.8$  is found to be almost constant ( $\cong 2$ ). We also note that the effective-mass ratio for the valence band with respect to UVB,  $m_5^*/m_1^*$ , is obtained to be 0.2, which in turn means that the carrier mobility in the valence band is extremely large (see Table II). Theoretical band calculations for these solid solutions will be required to confirm the above model.

## V. CONCLUSION

From the static and dynamic transport measurements for solid solutions of  $p$ -type  $\text{Sb}_{2-x}\text{In}_x\text{Te}_3$  ( $0 \leq x \leq 0.4$ )

and  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  ( $0 \leq y \leq 1.8$ ) along the  $C_2$  axis, we have found the following features. There are systemic variations of the Hall coefficient and Hall mobility with In content  $x$  and Se content  $y$  below  $y \leq 0.2$ . The most significant change in the band structure for  $\text{Sb}_2\text{Te}_{3-y}\text{Se}_y$  is the appearance of the valence band above  $y = 0.2$ , as found from the SdH effect and TTE experiments. The observed TTE voltages are characterized by a multiple relaxation process with a different relaxation time  $\tau_i$  ( $i = 1-6$ ) for thermal diffusions of photogenerated carriers, where  $\tau_1$  and  $\tau_2$  are due to holes in the UVB,  $\tau_3$  and  $\tau_4$  to LVB, and  $\tau_5$  and  $\tau_6$  (observable only for  $y \geq 0.7$ ) are due to the valence band with extremely large mobilities, which may originate from the valence bands of the constituent counterpart of  $\text{Sb}_2\text{Se}_3$ . Based on these experimental data, we have proposed the most probable band models for these solid solutions. The anisotropy in the effective mass in each band is evaluated, which indicates that the multivalley model is valid for the  $\text{Sb}_2\text{Te}_3$  based solid solutions.

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